

Hole Defects on Two-Dimensional Materials Formed by Electron Beam Irradiation: Toward Nanopore Devices

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Two-dimensional (2D) materials containing hole defects are a promising substitute for conventional nanopore membranes like silicon nitride. Hole defects on 2D materials, as atomically thin nanopores, have been used in nanopore devices, such as DNA sensor, gas sensor and purifier at lab-scale. For practical applications of 2D materials to nanopore devices, researches on characteristics of hole defects on graphene, hexagonal boron nitride and molybdenum disulfide have been conducted precisely using transmission electron microscope. Here, we summarized formation, features, structural preference and stability of hole defects on 2D materials with atomic-resolution transmission electron microscope images and theoretical calculations, emphasizing the future challenges in controlling the edge structures and stabilization of hole defects. Exploring the properties at the local structure of hole defects through in situ experiments is also the important issue for the fabrication of realistic 2D nanopore devices.

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INTRODUCTION

Defects on two-dimensional (2D) materials were often considered as hindrance despite of potential controllability in electromagnetic and mechanical property of the materials (Lahiri et al., 2010; Yazyev & Louie, 2010; Attacalite et al., 2011; Zhou et al., 2013a). For example, the actual performance of graphene devices failed to satisfy the expectations due to the inherent defect in them (Li et al., 2009). Although many efforts to remove the defects in 2D materials (Chen et al., 2013; Lam et al., 2014) have been performed, it has turned out to be only partial healing of the defects. Accordingly, people started to use these inevitable defects to many applications using the fact that defects can manipulate the electrical and magnetic properties of 2D materials. Among various types of defects, the hole defects that are made from ejection of atoms from the sheet were extensively studied for many

applications as atomically thin nanopores. In the case of graphene, many studies addressed possibility of hole defects as DNA sequencing (Garaj et al., 2010; Merchant et al., 2010; Schneider et al., 2010; Garaj et al., 2013), gas sensing (Dan et al., 2009), ion and molecules sieve (Sint et al., 2008; Koenig et al., 2012) and many other applications (Liu et al., 2012; Xu et al., 2013). Several comparable studies also have been done with hexagonal boron nitride (hBN) (Liu et al., 2013a) and molybdenum disulfide (MoS₂) (Farimani et al., 2014; Liu et al., 2014; Waduge et al., 2015). These studies focus on performance tests at the hole defects without the information of edge configuration and chemical stability which may affect to the performance significantly.

This paper summarizes the ways to produce holes by electron beam irradiation, control the size and shape and stabilize it for the representative 2D materials including graphene, hBN, and MoS₂. Although these three have hexagonal lattices, graphene

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consists of mono element of carbon, but hBN contains two elements of boron (B) and nitrogen (N) alternatively in plane, and each molybdenum and sulfur in MoS₂ are bound to each other with a ratio of 1:2 in a trigonal prism unit cell wherein Mo layer is sandwiched between sulfur layers (Fig. 1). Each material has its own composition, thus, it shows all different features such as formation process, edge structure, stability and consequential properties of hole defects on graphene, hBN, and MoS₂, which are described below.

Among various methods to make holes on 2D materials (Bieri et al., 2009; Girit et al., 2009; Bai et al., 2010; Kim et al., 2010; Koenig et al., 2012; Russo & Golovchenko, 2012), electron beam in transmission electron microscope (TEM) is good at size control at atomic scale, which is the most important issue for the sensitivity and selectivity of nanopore devices. If electron beam irradiation on a 2D specimen with a high electron energy breaks the atomic bonds within the material, which is called knock-on voltage, atoms are ejected from the lattice leaving holes on the materials. The atom displacement, knock-on thresholds and other structural information of graphene (Smith & Luzzi, 2001), hBN (Kotakoski et al., 2010), and MoS₂ (Komsa et al., 2012) are summarized in Table 1.

HOLE DEFECTS ON GRPAHENE

Fig. 2A shows the hole defects of graphene by the electron beam irradiation at 80 kV. To make hole defects in graphene, over 86 kV of electron beam energy, the knock-on threshold voltage of graphene, needs to be irradiated on the sample. But some studies showed the existence of oxygen or other chemicals on the sheet or inside TEM chamber can lower the knock-on threshold of graphene by chemical etching effect (Ramasse et al., 2012). Inherent defects created from the synthesizing process of graphene also lower the knock-on threshold voltage (Crespi et al., 1996). Once a vacancy is formed, it continuously grows as electron beam irradiates to make an enlarged hole in the sheet. The size of hole defect can be controlled by the time of electron beam irradiation at a given acceleration voltage.

The hole defect of graphene has mixed armchair and zigzag atomic configuration at the edge. Because the edge configuration of graphene are known to have significant influences to the electromagnetic property of graphene (Nakada et al., 1996; Son et al., 2006; Kim & Kim, 2008; Jung & MacDonald, 2009; Magda et al., 2014), the edge configuration and its stableness become one of major interests in studies of graphene hole defect. Direct atomic imaging at the graphene

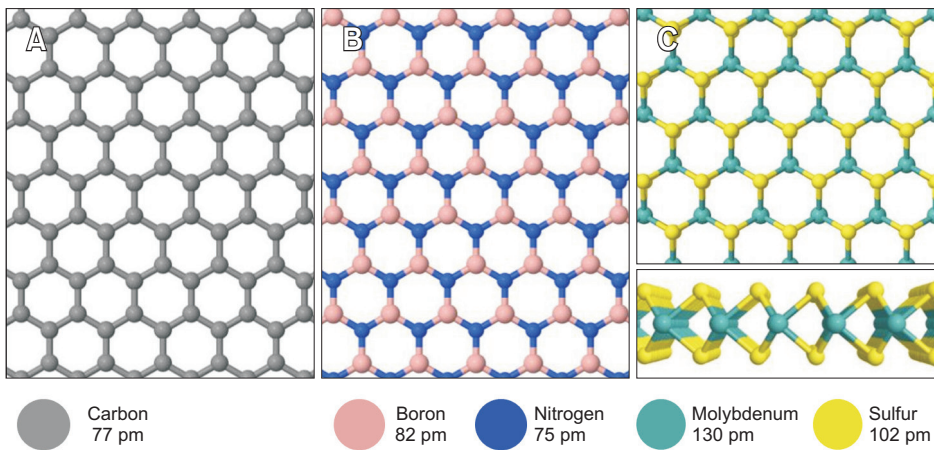


Fig. 1. Atomic models of monolayer graphene, hexagonal boron nitride, and molybdenum disulfide (2H) from a top and side view with covalent radii of atoms.

Table 1. Structural data of monolayer graphene, hBN, and MoS₂ (2H)

Reference	Material	Atomic number	Lattice system	Lattice parameter (Å)	Displacement threshold (eV)	Knock-on threshold (kV)
Smith & Luzzi (2001)	Graphene	C: 6	Hexagonal	2.46	17	86
Kotakoski et al. (2010)	hBN	B: 5	Hexagonal	2.51	B: 19	79.5
		N: 7			N: 23	118.6
Komsa et al. (2012)	MoS ₂ (2H)	S: 16	Hexagonal	3.2	S: 6.9	80
		Mo: 42			Mo: 20	560

hBN, hexagonal boron nitride; MoS₂, molybdenum disulfide.

The calculated values of displacement threshold and knock-on threshold of graphene, hBN, and MoS₂ are taken from Smith & Luzzi (2001) (*J. Appl. Phys.* 90, 3509-3515); Kotakoski et al. (2010) (*Phys. Rev. B* 82, 113404); Komsa et al. (2012) (*Phys. Rev. Lett.* 109, 035503), respectively.

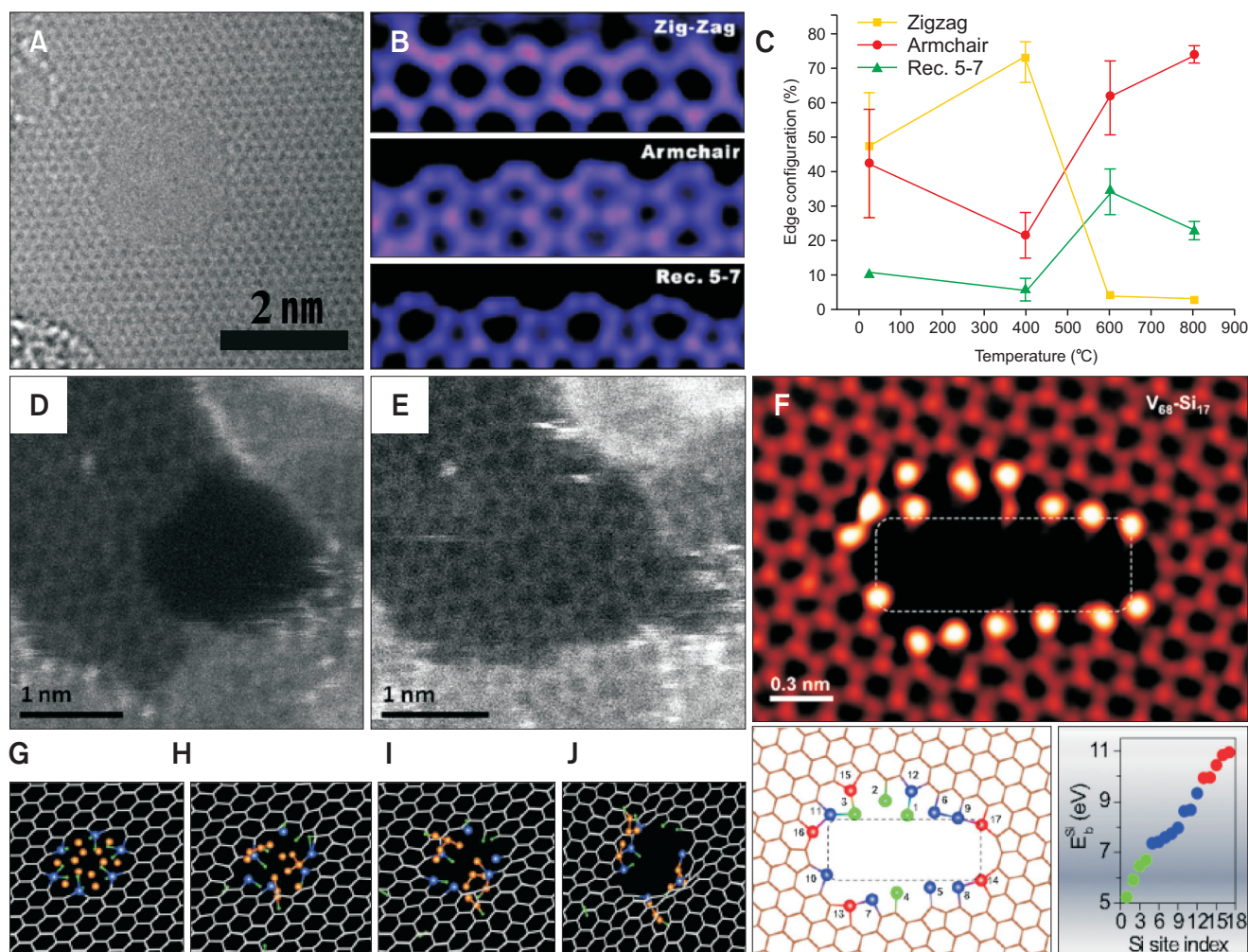


Fig. 2. Features and stability of graphene hole defects. (A) Aberration-corrected transmission electron microscope (TEM) image of graphene hole defect. (B) Graphene has zigzag, armchair, reconstructed (Rec.) 5 to 7 zigzag configuration at the edges. (C) Percentage of each edge configurations is marked on the graph. It shows temperature dependence of edge configurations. (D) Scanning TEM (STEM) image of hole defect created near the hydrocarbon contamination. (E) The hole is filled with C polygons even under ultra high vacuum condition. (F) STEM image and structure model of Si-passivated graphene hole defect and binding energy of Si atoms at the edge. Si-passivated hole defect is stable against hole filling. (G-J) ab initio molecular dynamics simulations reveal the Si-passivated hole forms bonding with C adatoms out of the graphene plane, preventing the hole filling. Fig. 2B and C are reprinted from He et al. (2015) (*ACS Nano* 9, 4786-4795) with original copyright holder's permission. Fig. 2D and E are reprinted from Zan et al. (2012) (*Nano Lett.* 12, 3936-3940) with original copyright holder's permission. Fig. 2F-J are reprinted from Lee et al. (2014) (*Proc. Natl. Acad. Sci.* 111, 7522-7526) with original copyright holder's permission.

edges using aberration-corrected TEM gave the insight on the stability of edge configurations (Girit et al., 2009; Song et al., 2011; Kim et al., 2013a; He et al., 2015). Recently, He et al. (2015) reported the temperature dependence of graphene edge configuration through in situ heating experiment using aberration-corrected TEM. As shown in Fig. 2B and C, armchair and reconstructed 5 to 7 zigzag configurations were predominant above 600°C which has good agreement with the theoretical predictions (Koskinen et al., 2008). Below 400°C, zigzag configuration was dominantly observed, which was led by major involvement of chemical etching process from contamination of sample. It gives a prospect in edge

control of graphene hole defect for graphene nanopore devices, though contamination effect still remains an issue to be resolved.

Meanwhile, high reactivity of the edges of graphene hole defect has been a big obstacle to realization of graphene nanopore devices. Graphene has high reactivity with chemicals, especially at the hole edges that have dangling bonds. For example, DNA would be fond of sticking to the hole edges and surface when translocating the graphene nanopore, which makes it complicate to identify the DNA sequence (Sathe et al., 2011; Wells et al., 2012). Moreover, despite all the efforts to make holes, small holes in graphene are filled with

carbon adatoms nearby within hours even under ultra-high vacuum conditions of TEM as shown in Fig. 2D and E (Zan et al., 2012). Thus, fixation or stabilization of hole defects is emerging as a key issue for graphene nanopore devices and passivation of hole edges with other atoms was suggested as one potential method as shown in Fig. 2F-J (Lee et al., 2014). Si-passivated holes in graphene were directly observed using scanning TEM, and proved to be stable against carbon filling even under intense electron beam condition and ambient atmospheric condition. Molecular dynamics simulations supported this observation, showing that carbon adatoms would stick out of the graphene plane preventing filling hole defects. Si passivation opens potential of stabilization of hole defects, however, the fabrication methods should be further developed. Additionally, considering the binding energy between C-Si at armchair site is higher than that at zigzag site, importance of understanding edge configuration of graphene hole defects is highlighted again. Therefore, in-depth studies on the edge configuration and stabilization will make progress toward graphene nanopore devices.

HOLE DEFECTS ON hBN

Compared to graphene, hBN has received less attention as a 2D device due to the difficulties in getting synthesized large sheet and non-conducting property (Kim et al., 2013b). As an insulator, hBN has been used as a substrate of graphene to show high electrical performance (Dean et al., 2010), but not much used alone. However, Liu et al. (2013a) remarked hBN may exhibit superior durability and insulating properties in high-ionic strength solution compared with graphene and realized hBN nanopore device for DNA sensor.

The notable thing is that hBN hole defects have a unique feature in shape, which is controllable unlike graphene. Fig. 3A-C represent a triangular hole defect in monolayer hBN sheet. Because boron has a lower knock-on threshold than nitrogen under transmission electron beam (Table 1), 80 kV of electron beam preferentially knock off the boron atoms first, making B monovacancies with N terminated edge along the hole defect edge (Jin et al., 2009; Meyer et al., 2009; Kotakoski et al., 2010; Ryu et al., 2015) as shown in Fig. 3A.

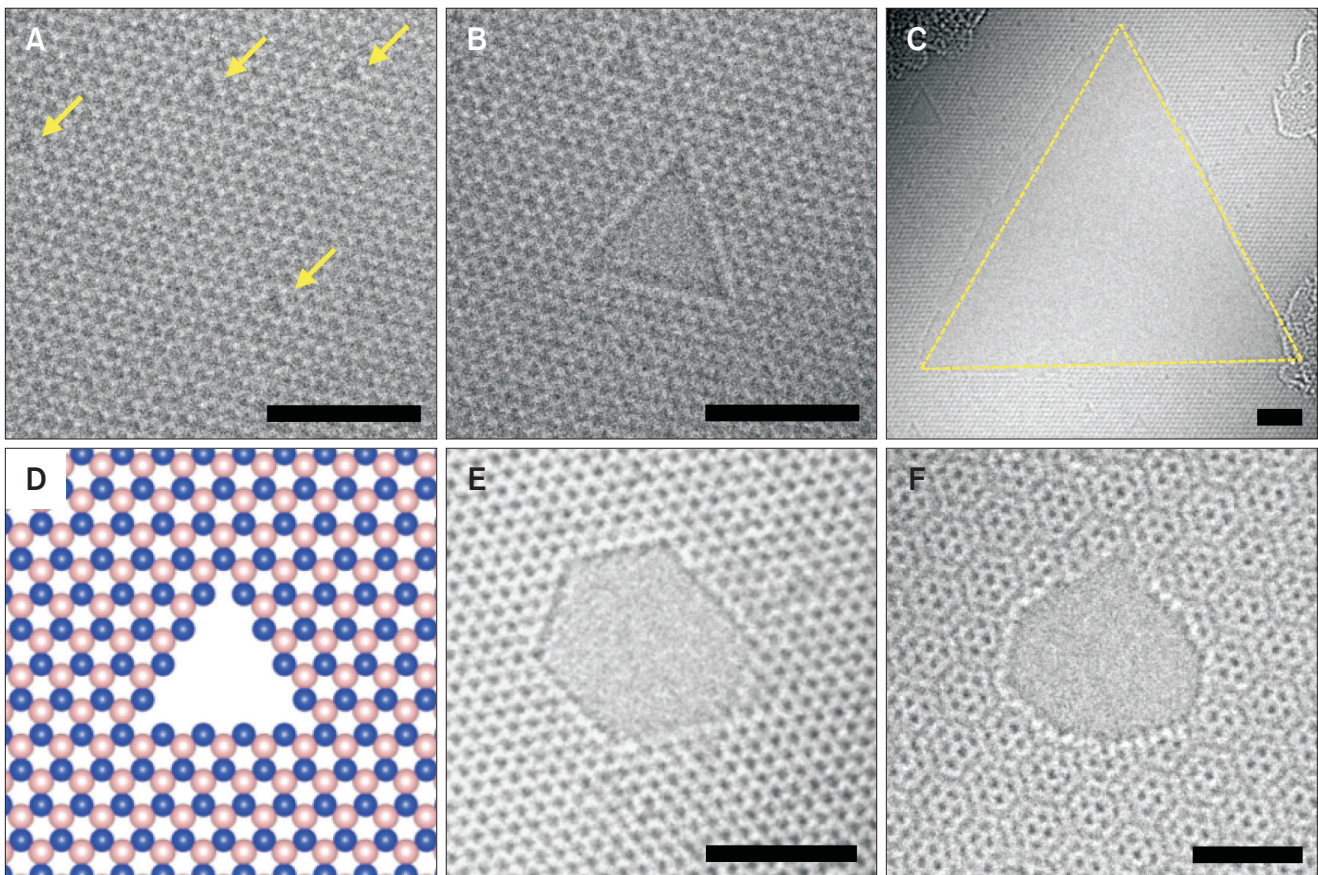


Fig. 3. Hole defects on mono- and double-layer hexagonal boron nitride (hBN). (A-C) In monolayer hBN, hole defects grow with maintaining triangular shape from monovacancy pointed out by yellow arrows (A) to enlarged ($\sim 110 \text{ nm}^2$) hole (C) by electron beam irradiation. (D) Atomic model of N-terminated edges of hBN hole defect. The pink and blue balls represent boron and nitrogen atoms, respectively. (E) Hexagonal shape of hole defect is created on AA' stacked hBN double layer. (F) Randomly shaped hole defect is created on rotated hBN double layer. Scale bars=2 nm.

Fig. 3B, the hole defect formed by prolonged electron beam irradiation at Fig. 3A is defined as triangular shape more clearly. This triangular hole defect maintains its shape as it grows which is verified up to area of 110 nm^2 as shown in Fig. 3C. Atomic model of triangular hole defect in Fig. 3D shows that the edges are terminated with N atoms. According to the calculation result of Kotakoski et al. (2010), N monovacancy as well as B monovacancy is expected to be formed by above 120 kV of electron beam. Nevertheless, all experimental results using TEM have shown N terminated triangular hole defect starting from B monovacancy. Yin et al. (2010) explained by cohesive energy calculations that N-terminated hole defect is more stable than B-terminated one due to the ionic character of the material. In our previous work, we found out N-terminated triangular holes are formed even at the edge of the sheet not only inside of the sheet and merging of two holes also results in the restoration of triangular shape (Ryu et al., 2015). Consequently, hole defects created by electron beam on monolayer hBN sheet are always featured by triangular holes with N terminated edges. Also note that the edges have solely zigzag configuration in case of formation by electron beam, which is another uniqueness compared to the edges of graphene hole defects that have mixed zigzag and armchair configuration.

Furthermore, hexagonal hole defect could be fabricated in AA' stacked double layer hBN sheet like Fig. 3E. The nature of AA' stacking structure of hBN, which B atoms are on top of N atoms and vice versa, makes the orientation of defects in one layer 180 degree opposite to those in the other layer of hBN sheet (Meyer et al., 2009). Two opposite triangle defects grow and result in a hexagonal hole defect with N terminated zigzag edges. Because AA' stacking structure is known to be the most stable structure in bulk hBN structure (Alem et al., 2009), AA' stacked double layer hBN can be easily obtained through the scotch-tape exfoliation (Pacile et al., 2008). In Fig.

3F, if double layer of hBN is randomly rotated each other by stacking two monolayer of hBN, hole defects have no longer triangular or hexagonal shaped hole defects. The rotated stacking structure can be directly inferred from the Moire pattern. To sum up, ultimately we are able to manufacture freely the hBN nanopores in shape by managing the number of layer and stacking structure, and in size by the irradiation time. Due to these shape and size of hBN hole defects with controllable manners, we are expecting the utilization of the hBN nanopores as a nano-patterning template. Also N-terminated hole defect of hBN is found to give enhanced half-metallicity and large magnetism. It suggests the potential of hBN nanopore application for spintronics, light emission and photocatalysis (Du et al., 2009).

HOLE DEFECTS ON MoS_2

Recently, MoS_2 has been growing interest as a promising material due to its semiconducting nature which makes it facile to make sensing and electronic applications (Farimani et al., 2014; Liu et al., 2014; Waduge et al., 2015). Especially as a DNA detecting sensor, MoS_2 nanopore membranes have shown better performance than graphene nanopore membranes for transverse detection without special surface treatment process to prevent the interaction between DNA and the surface, unlike graphene (Wells et al., 2012).

MoS_2 is commonly found in 2H form in nature among three polytypes: 1T, 2H, and 3R. Synthesized MoS_2 films may have 3R structures (Wilson & Yoffe, 1969) but most of studies on MoS_2 defects are performed with exfoliated MoS_2 , so with 2H structure. The top and side views of 2H MoS_2 are shown in Fig. 1C. The smallest hole in 2H MoS_2 can be created by the removal of one Mo atom or two S atoms but much higher concentration of S site holes than Mo site holes are found in TEM due to the different knock-on threshold of S and Mo

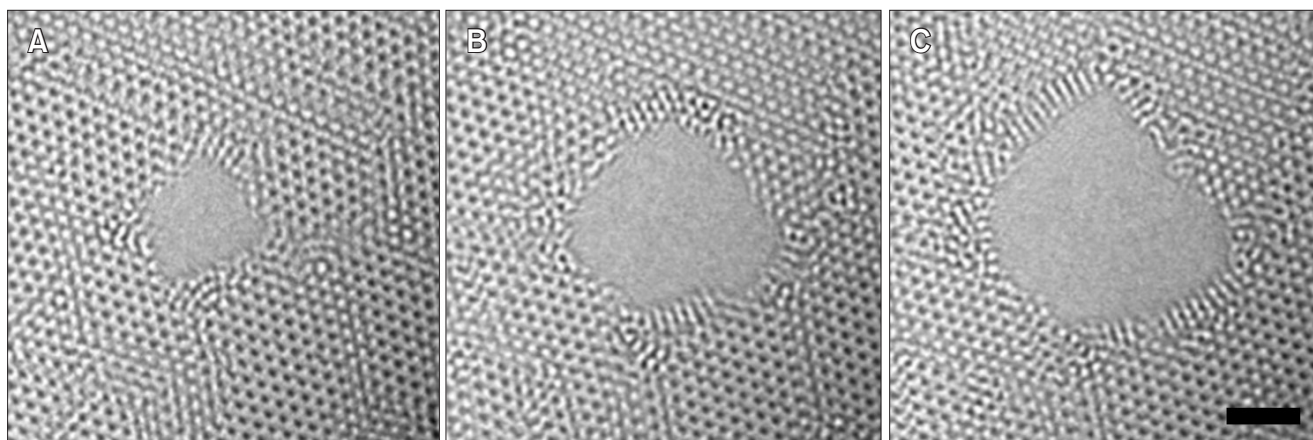


Fig. 4. The sequential growth process of a hole defect on molybdenum disulfide (MoS_2) by electron beam irradiation in monolayer MoS_2 sheet. Mo atoms aggregate at the edge. Scale bar=2 nm.

atoms (Hong et al., 2015).

Fig. 4 is the serial images of MoS₂ hole defect created by 80 kV of electron beam in TEM. Because the knock-on thresholds of S and Mo atom are 80 kV and 560 kV respectively as shown in Table 1, S vacancies are formed first by slightly focused electron beams. Meanwhile Mo atoms are less likely to be ejected by 80 kV electron beam, resulting in the agglomeration of Mo atoms at the edge of hole defects. From Fig. 4, Mo atoms increasingly aggregate at the edge under continued electron beam irradiation. Although a few papers reported the phenomenon of Mo atoms agglomeration (Liu et al., 2013b; Zan et al., 2013), none reported the corresponding effects in electrical or magnetic properties at the edge. According to the theoretical calculations (Zhou et al., 2013b), Mo-Mo metallic bonds are formed at S vacancy sites and cancel the magnetism by pairing the unsaturated spin electrons. Similarly, the degree of Mo atoms agglomeration may affect to the metallicity so to the electrical performance such as the current signals when DNA or molecules transverse the holes. Therefore, for MoS₂-based nanopore devices, understanding of edge configuration of MoS₂ hole defects and consequent electrical and magnetic properties is required.

CONCLUSIONS

Hole defects on 2D materials have been studied extensively from fabrication, features, control to stabilization. In summary, graphene hole defects have mixed armchair, zigzag and reconstructed 5 to 7 zigzag configurations at the edge which are temperature-dependent. Due to the high reactivity of graphene and its hole defect with chemicals, additional

process of stabilization is required. hBN hole defects have advantage in shape control to triangle, hexagon and randomly shaped hole by managing the number of layers and the stacking structure. And it always has N-terminated zigzag edges which make it possible to control the electromagnetic properties for nanopore devices. MoS₂ hole defects show Mo atoms agglomeration at the edges. Based on these studies, 2D materials nanopores become more realizable. TEM experiments have made a significant contribution to understanding of hole defects on 2D materials by direct imaging of the features and dynamics of hole defects; however, interconnecting the structure and property of hole defects was lack. In situ experiments using the real-time imaging and performance measurement in TEM would accelerate the realization of 2D materials-based nanopore devices.

EXPERIMENTAL METHODS

Some images except reprinted ones with permission from references in figures were taken by the authors of this paper using an aberration-corrected FEI Titan Cube TEM (FEI Titan³ G2 60-300; FEI, Netherlands) with a monochromator at 80 kV. Imaging condition was settled as $-21 \pm 0.5 \mu\text{m}$ of spherical aberration (Cs), around $5 \times 10^5 \text{ e}^- \text{ nm}^{-2}$ of electron beam densities. Images were taken with 0.5 seconds of exposure time.

CONFLICT OF INTEREST

No potential conflict of interest relevant to this article was reported.

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