# Rational design of gold catalysts with enhanced thermal stability: post modification of $Au/TiO_2$ by amorphous SiO<sub>2</sub> decoration

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 $Au/TiO_2$  is highly active for CO oxidation, but it often suffers from sintering in high-temperature environments. In this work, we report on a novel design of gold catalysts, in which pre-formed Au/TiO<sub>2</sub> catalysts were post decorated by amorphous SiO<sub>2</sub> to suppress the agglomeration of gold particles. Even after being aged in O<sub>2</sub>-He at 700 °C, the SiO<sub>2</sub>-decorated Au/TiO<sub>2</sub> was still active for CO oxidation at ambient temperature.

KEY WORDS: gold nanoparticles; CO oxidation; promotion; sintering; alkoxysilanes; post decoration.

### 1. Introduction

One of the driving forces of contemporary catalysis research is the rational design of novel catalytic materials [1,2]. In rational design, one first identifies the key functions to be actualized and then advertently synthesizes catalytic materials according to the desired functions [3–5]. In the specific case of catalysis by supported gold nanoparticles, two functionalities, namely, the catalytic activity and thermal stability of catalysts, are particularly important. It is known that Au/SiO<sub>2</sub> usually exhibits very low activity in CO oxidation. On the other hand,  $Au/TiO_2$  is highly active for CO oxidation, but it suffers from significant loss of activity upon high-temperature calcination [6–10]. The latter situation may adversely influence the practical applications of Au/ TiO<sub>2</sub> when the operation and regeneration of catalysts at elevated temperature are needed [11,12].

To improve the activity of SiO<sub>2</sub>-based gold catalysts, our group has modified mesoporous [13] or fumed [14] SiO<sub>2</sub> supports with TiO<sub>2</sub> coatings via a surface-sol-gel method, and then loaded gold. The idea is to increase the isoelectric point of the support and to control the local structure of active centers through surface functionalization so as to facilitate the deposition of gold and to enhance the catalytic activity. Other researchers have modified SiO<sub>2</sub> supports by TiO<sub>2</sub> [15,16] or CoO<sub>x</sub> [17–19] for loading gold. Alternatively, we used Au(en)<sub>2</sub>Cl<sub>3</sub> instead of HAuCl<sub>4</sub> as the precursor, and succeeded in the preparation of highly active Au/SiO<sub>2</sub> catalysts through deposition of the above gold precursor under basic conditions [20,21]. Other researchers have

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also successfully prepared active  $Au/SiO_2$  with different designs [22–25].

To improve the thermal stability of Au/TiO<sub>2</sub>, we coated TiO<sub>2</sub> with a thin layer of Al<sub>2</sub>O<sub>3</sub> via a surface-solgel method, and then loaded gold nanoparticles [26]. The maintenance of activity in CO oxidation after calcination at 500 °C was ascribed to the stabilization of gold particles against agglomeration [26]. Others have loaded gold particles on MgO- or BaO-modified Al<sub>2</sub>O<sub>3</sub>, and found similar stabilization effects [27,28]. Recently, we demonstrated that the surface-sol-gel method [26] is not necessary because conventional impregnation using Al(NO<sub>3</sub>)<sub>3</sub> precursor followed by thermal decomposition to form dispersed Al<sub>2</sub>O<sub>3</sub> on TiO<sub>2</sub> worked well, and that Al<sub>2</sub>O<sub>3</sub> additive [26] is not unique because other selected metal oxide additives exhibited comparable or even better stabilizing effects [29].

In this work, we report on a rational design of Au/ TiO<sub>2</sub>-based catalysts with enhanced thermal stability through post modification of Au/TiO<sub>2</sub> by amorphous SiO<sub>2</sub> decoration (figure 1). This design is essentially different from the conventional one in which surfacemodified oxide supports were prepared first, and then gold was loaded [13–16,26–29]. The current post-decoration design is based on the rationale that the preformed gold-TiO<sub>2</sub> interface may provide high activity for CO oxidation [6–10], whereas the subsequently loaded SiO<sub>2</sub> decoration may serve as a textural promoter to mitigate the agglomeration of gold nanoparticles.

## 2. Experimental

Au/TiO<sub>2</sub> was prepared by deposition-precipitation [6–10]. First, 150 mL aqueous  $HAuCl_4$  solution



Figure 1. Proposed scheme for the synthesis of SiO<sub>2</sub>-decorated Au/TiO<sub>2</sub>. Pre-formed Au/TiO<sub>2</sub> was treated in tetramethyl orthosilane (TMOS, Si(OCH<sub>3</sub>)<sub>4</sub>), (3-aminopropyl)trimethoxysilane (APTMS,  $H_2N(CH_2)_3Si(OC_2H_5)_3$ ), or tris(*tert*-butoxy)silanol (BOS, ((CH<sub>3</sub>)<sub>3</sub>CO)<sub>3</sub>SiOH) solution, followed by calcination.

containing 0.9 g HAuCl<sub>4</sub>·3H<sub>2</sub>O was titrated by 5.0 wt% NaOH to pH 10, and mixed with 4.0 g TiO<sub>2</sub> (Degussa P25, surface area 48 m<sup>2</sup>/g) under vigorous stirring. The suspension was subsequently re-titrated to pH 10, and stirred at 70 °C for 2 h. The precipitate was centrifuged, thoroughly washed with water, dried in a vacuum oven at 60 °C overnight, and calcined in a muffle oven at 300 °C for 3 h.

To synthesize SiO<sub>2</sub>-decorated Au/TiO<sub>2</sub> (denoted as SiO<sub>2</sub>/Au/TiO<sub>2</sub>), 0.5 g Au/TiO<sub>2</sub> (which was calcined in a muffle oven at 300 °C for 3 h, as described above) was dispersed in 100 mL ethanol, 3.4 g tetramethyl orthosilane (TMOS, Si(OCH<sub>3</sub>)<sub>4</sub>), or 1.0 g (3-aminopropyl)trimethoxysilane (APTMS, H<sub>2</sub>N(CH<sub>2</sub>)<sub>3</sub>Si(OC<sub>2</sub>H<sub>5</sub>)<sub>3</sub>), or 0.5 g tris(*tert*-butoxy)silanol (BOS, ((CH<sub>3</sub>)<sub>3</sub>CO)<sub>3</sub>SiOH) added, and the suspension stirred at room temperature for 5–15 h under N<sub>2</sub> atmosphere. The product was centrifuged and dried in a vacuum oven at 60 °C.

To synthesize gold supported on SiO<sub>2</sub>-decorated TiO<sub>2</sub> (denoted as Au/SiO<sub>2</sub>/TiO<sub>2</sub>) for a control experiment, 1.0 g TiO<sub>2</sub> was dispersed in 200 mL ethanol, 6.8 g tetramethyl orthosilane added, and the suspension stirred at room temperature for 15 h. The resulting material was centrifuged, dried in vacuum, and calcined in a muffle oven at 500 °C for 2 h. Gold was loaded via deposition-precipitation according to the procedure described in the above paragraph, and the resulting sample was calcined in a muffle oven at 300 °C for 3 h.

To perform reaction testing, 50 mg catalyst was packed into a U-shaped silica tube (4 mm i.d.), and was calcined under flowing 8% O<sub>2</sub>–He at 500 or 700°C for 1–5 h (heating rate: 30 °C/min) on an AMI 200 microreactor. CO oxidation was performed by flowing 1% CO (balance air, <4 ppm H<sub>2</sub>O) through the catalyst at

a rate of 37 cm<sup>3</sup>/min, and the reaction temperature was varied using a furnace or by immersing the U-type tube in ice-water or acetone-liquid  $N_2$ . The reaction mixture was analyzed by a dual-column GC with a thermal conductivity detector.

To characterize the catalysts, XRD data were collected on a Siemens D5005 diffractometer with Cu K<sub> $\alpha$ </sub> radiation. The average gold particle sizes were estimated from X-ray line broadening analysis applying the Debye-Scherrer equation on the (220) diffraction ( $2\theta = 44^{\circ}$ ) of gold. The gold content was analyzed using inductivity coupled plasma-optical emission spectrometry on a Thermo IRIS Intrepid II spectrometer. The Si/Ti molar ratio was estimated by EDAX on a JOEL JSM-6060 scanning electron microscope coupled with an EDAX detector. TGA/DTG experiments were conducted on a TGA 2950 instrument using a heating rate of 10 °C/min under air atmosphere. TEM experiments were performed using a Hitachi HD-2000 STEM operating at 200 kV.

#### 3. Results and discussion

### 3.1. $Au/TiO_2$ : activity and thermal sintering

Figure 2A shows the light-off curves of Au/TiO<sub>2</sub>. Au/ TiO<sub>2</sub>, calcined in a muffle oven at 300 °C according to the temperature used by Haruta *et al.* [6], was very active for CO oxidation, achieving T<sub>50</sub> (temperature required for 50% conversion) value of -37 °C. However, its catalytic activity significantly decreased upon further calcination at higher temperatures in flowing O<sub>2</sub>-He. The T<sub>50</sub> values increased to 30, 84, and 127 °C when Au/TiO<sub>2</sub> was calcined at 500 °C for 1, 3, and 5 h,



Figure 2. CO light-off curves (A) and XRD patterns (B) of Au/TiO<sub>2</sub> calcined in a muffle oven at 300 °C and Au/TiO<sub>2</sub> further calcined in O<sub>2</sub>-He at 500 °C for 1–5 h or at 700 °C for 1 h.

respectively, and further increased to 196 °C when Au/ $TiO_2$  was calcined at 700 °C for 1 h, indicating severe thermal deactivation commonly seen in the literature [26,29–31].

Figure 2B compares the XRD patterns of Au/TiO<sub>2</sub> after different treatments. In general, gold peaks appeared at  $2\theta = 38, 44, 65, \text{ and } 78^\circ$ , corresponding to Au (111), (200), (220), and (311), respectively [31]. Other peaks are ascribed to those from anatase and rutile, characteristic of Degussa P25 [32]. The gold peaks became sharper with the increases in the duration and temperature of calcination, implying the growth of gold particles. The average gold particle size of Au/TiO<sub>2</sub> (300 °C) was estimated as 4.0 nm according to X-ray line broadening analysis. The value increased to 6.2, 7.5, and 9.5 nm after Au/TiO2 was calcined at 500 °C for 1, 3, and 5 h, and increased to 16.5 nm after it was calcined at 700 °C for 1 h (table 1), meaning severe agglomeration of gold particles. The increase in gold particle size (table 1) is consistent with the decrease in catalytic activity (figure 2A).

In principle, metal nanoparticles have extra surface free energy and higher chemical potential than those of the bulk metal, so that they can be highly mobile even below the melting point of the bulk metal [10]. In fact, the melting point of gold nanoparticles decreases with the particle size, and is much lower than that of the bulk gold [33]. The sintering problem of gold nanaparticles is particularly acute if they are placed in  $O_2$  atmosphere as compared to vacuum [34,35]. Because the optimal size of gold particles on TiO<sub>2</sub> for CO oxidation is 3–5 nm and the activity decreased greatly when the gold particle become too big [6–10,36], the agglomeration of gold particles upon high-temperature calcination is the reason that may cause catalyst deactivation.

# 3.2. SiO<sub>2</sub>(TMOS)|Au|TiO<sub>2</sub>: activity and thermal sintering

To mitigate the sintering of gold nanoparticles, we modified the pre-formed Au/TiO<sub>2</sub> by SiO<sub>2</sub> adlayers via chemical grafting (figure 1). Figure 3A shows the CO light-off curves on SiO2-decorated Au/TiO2 prepared using Si(OCH<sub>3</sub>)<sub>4</sub> (TMOS) [37,38] as the precursor. It should be mentioned that the as-synthesized SiO<sub>2</sub>(T-MOS)/Au/TiO<sub>2</sub> was not very active for CO oxidation, achieving  $T_{50}$  at 239 °C, in sharp contrast to the -37 °C value obtained with Au/TiO<sub>2</sub> calcined at 300 °C, even though as-synthesized and 300 °C-calcined samples had similar gold particle sizes (table 1). The ineffectiveness of the as-synthesized sample may be due to the presence of unburned organic fragments, as implied from our TG/DTG experiments (figure 4). Other researchers have found that gold catalysts containing certain organic ligands or fragments are not active unless these are removed through proper pretreatment [25,31,39-45].

Once the SiO<sub>2</sub> precursor was sufficiently decomposed and SiO<sub>2</sub> formed, the sample exhibited improved thermal stability. The T<sub>50</sub> values of SiO<sub>2</sub>(TMOS)/Au/TiO<sub>2</sub> were 27, 50, and 57 °C when the sample was calcined in O<sub>2</sub>-He at 500 °C for 1, 3, and 5 h, respectively. The T<sub>50</sub> value was maintained at 64 °C when SiO<sub>2</sub>(TMOS)/Au/ TiO<sub>2</sub> was subject to calcination at 700 °C (figure 3 and table 1). Recall that the T<sub>50</sub> of unmodified Au/TiO<sub>2</sub> was as large as 196 °C after aging at 700 °C (figure 2 and table 1). Thus, the stabilizing effect of the SiO<sub>2</sub> adlayer is obvious.

Figure 3B shows the XRD patterns of SiO<sub>2</sub>-decorated Au/TiO<sub>2</sub> after different treatments. The gold peaks were all broad, and no drastic sharpening of the gold peaks was seen after thermal treatments. In particular, the average gold particle size of SiO<sub>2</sub>(TMOS)/Au/TiO<sub>2</sub> was estimated as 6.5 nm after 700 °C-calcination, much smaller than the 16.5 nm value with unmodified Au/ TiO<sub>2</sub> after 700 °C-calcination (table 1). Preliminary TEM data indicated that 700 °C-calcined SiO<sub>2</sub>(TMOS)/ Au/TiO<sub>2</sub> mainly contained gold particles with sizes of 3–15 nm, and 700 °C-calcined Au/TiO<sub>2</sub> mainly contained gold particles with sizes of 10–50 nm (data not shown). These data further confirm the stabilizing effect of the amorphous SiO<sub>2</sub> modifier.

Another noticeable trend in figure 3B is that the relative intensities of  $TiO_2$  peaks were not significantly

Table 1

Gold loadings (measured by ICP), Si/Ti ratios (by EDAX), average gold particle sizes (by XRD), and T <sub>50</sub> values of Au/TiO <sub>2</sub> , SiO <sub>2</sub> /Au/TiO <sub>2</sub> , and Au/SiO <sub>2</sub> /TiO <sub>2</sub> samples							
Sample	Au loading, wt%	Si/Ti molar ratio	Average gold particle size <sup>a</sup> , nm	T <sub>50</sub> , °C			
Au/TiO <sub>2</sub> (300 °C, muffle oven)	3.3		4.0	-37			

Au/TiO <sub>2</sub> (300 °C, muffle oven)	3.3		4.0	-37
Au/TiO <sub>2</sub> -C500-1 h			6.2	30
Au/TiO <sub>2</sub> -C500-3 h			7.5	84
Au/TiO <sub>2</sub> -C500-5 h			9.5	127
Au/TiO <sub>2</sub> -C700-1 h			16.5	196
As-synthesized SiO <sub>2</sub> (TMOS)/Au/TiO <sub>2</sub>			4.5 <sup>b</sup>	239
SiO <sub>2</sub> (TMOS)/Au/TiO <sub>2</sub> -C500-1 h	2.2	0.10	5.3	27
SiO <sub>2</sub> (TMOS)/Au/TiO <sub>2</sub> -C500-3 h			5.5	50
SiO <sub>2</sub> (TMOS)/Au/TiO <sub>2</sub> -C500-5 h			6.0	57
SiO <sub>2</sub> (TMOS)/Au/TiO <sub>2</sub> -C700-1 h			6.5	64
As-synthesized SiO <sub>2</sub> (APTMS)/Au/TiO <sub>2</sub>			4.7 <sup>b</sup>	228
SiO <sub>2</sub> (APTMS)/Au/TiO <sub>2</sub> -C500-1 h	2.5	0.06	5.4	26
SiO <sub>2</sub> (APTMS)/Au/TiO <sub>2</sub> -C500-3 h			5.7	30
SiO <sub>2</sub> (APTMS)/Au/TiO <sub>2</sub> -C500-5 h			6.2	29
SiO <sub>2</sub> (APTMS)/Au/TiO <sub>2</sub> -C700-1 h			7.1	85
As-synthesized SiO <sub>2</sub> (BOS)/Au/TiO <sub>2</sub>			4.7 <sup>b</sup>	141
SiO <sub>2</sub> (BOS)/Au/TiO <sub>2</sub> -C500-1 h	2.5	0.004	5.3	15
SiO <sub>2</sub> (BOS)/Au/TiO <sub>2</sub> -C500-3 h			5.9	40
SiO <sub>2</sub> (BOS)/Au/TiO <sub>2</sub> -C500-5 h			7.0	69
SiO <sub>2</sub> (BOS)/Au/TiO <sub>2</sub> -C700-1 h			12.0	110
Au/SiO <sub>2</sub> (TMOS)/TiO <sub>2</sub> (300 °C, muffle oven)			b, c	160
Au/SiO <sub>2</sub> (TMOS)/TiO <sub>2</sub> -C700-1 h	0.1	0.11	с	> 520

<sup>a</sup> measured after reaction testing unless otherwise indicated.

<sup>b</sup> measured before reaction testing.

<sup>c</sup> gold peaks were not clearly detected by XRD because of low gold loading.

changed. This is in contrast to the case shown in figure 2B, in which the peaks corresponding to the rutile phase ( $2\theta = 27$ , 36, 41, and 54 °) obviously intensified after calcining Au/TiO<sub>2</sub> at 700 °C. By comparing both figures 2B and 3B, it is clear that the post-decoration of Au/TiO<sub>2</sub> by SiO<sub>2</sub> not only suppressed the growth of gold, but also suppressed the transformation of anatase to rutile. It is well known in the literature that the dispersion of sulfate, like that of SiO<sub>2</sub> in the current case, can retard the phase transformation of TiO<sub>2</sub> [46]. It is also expected that certain structural changes of the supports during calcination may bring detrimental effects to the stabilization of gold particles [10].

Although the effects of  $SiO_2$  on catalytic activity and structure of the catalyst are obvious (figure 3), no crystalline  $SiO_2$  peaks were detected by XRD in figure 3B. This could be potentially interpreted as either the  $SiO_2$  species was amorphous, or the uptake of  $SiO_2$ was negligible. However, Si element was clearly detected in EDAX experiments (figure 5), and the Si/Ti molar ratio was estimated as 0.1, thus confirming that  $SiO_2$ was loaded. In fact, it is well established that alkoxysilanes can condense with hydroxyl groups on metal oxide surfaces [37,38].

# 3.3. Effect of other $SiO_2$ precursors

To test whether the methodology of decorating  $Au/TiO_2$  by amorphous  $SiO_2$  is general, other  $SiO_2$  precur-

sors, such as  $H_2N(CH_2)_3Si(OC_2H_5)_3$  (APTMS) [47] and ((CH<sub>3</sub>)<sub>3</sub>CO)<sub>3</sub>SiOH (BOS) [48] were preliminarily tried. The amine group of APTMS is known to interact with gold surfaces [49], and the  $OC_2H_5$  groups can also condense with the surface hydroxyl groups [47]. This amine-gold interaction may provide a driving force to enrich APTMS in the close vicinity of gold nanoparticles, thereby controlling the spatial location of the surface decoration of  $SiO_2$  (figure 1). On the other hand, the rationale behind the use of BOS is to investigate the use of the stereochemistry of siloxane precursors in controlling the surface decoration density of SiO<sub>2</sub> (figure 1). As seen from table 1, the Si content of  $SiO_2(APTMS)/Au/TiO_2$  (Si/Ti ~ 0.06) is only a little lower than that of SiO<sub>2</sub>(TMOS)/Au/TiO<sub>2</sub> (Si/Ti  $\sim 0.1$ ), while the Si content of SiO<sub>2</sub>/(BOS)/Au/TiO<sub>2</sub> (Si/Ti  $\sim$ 0.004) is the lowest. The low Si content of  $SiO_2/(BOS)/$  $Au/TiO_2$  could be attributed to the fact that BOS is bulkier than TMOS, thus resulting in the low surface functonalization efficiency of BOS (figure 1). Alternatively, one could also argue that this may be due to the fact that the amount of BOS (0.5 g, equivalent to 0.002 mol) put in the synthesis mixture was less than that of TMOS (3.4 g, equivalent to 0.02 mol). However, preliminary data obtained by our group indicated that when the chemical grafting was carried out room temperature, as the grafting temperature used here, even using much more BOS, the Si/Ti ratio was still comparably low [50].



Figure 3. CO light-off curves (figure 3A) and XRD patterns (figure 3B) of as-synthesized SiO<sub>2</sub>(TMOS)/Au/TiO<sub>2</sub> and those further calcined in O<sub>2</sub>-He at 500 °C for 1–5 h or at 700 °C for 1 h.

The as-synthesized SiO<sub>2</sub>(APTMS)/Au/TiO<sub>2</sub> and SiO<sub>2</sub>(BOS)/Au/TiO<sub>2</sub> were again not very active, achieving 50% CO conversion at 228 and 141 °C, respectively (table 1). This is again due to the presence of residual organic fragments, as verified from TG/DTG experiments similar to the one in figure 4 (data not shown). The residual organic fragments do not have to be completely removed to actualize some activity,



Figure 4. TG/DTG data of SiO<sub>2</sub>(TMOS)/Au/TiO<sub>2</sub>. The experiment, starting from as-synthesized SiO<sub>2</sub>(TMOS)/Au/TiO<sub>2</sub>, was conducted with a heating rate of 10  $^{\circ}$ C/min in air.



Figure 5. EDAX data of SiO<sub>2</sub>(TMOS)/Au/TiO<sub>2</sub> which was calcined in O<sub>2</sub>–He at 500 °C.

because the overall activity is expected to depend on the coverage of surface fragments and the population of the uncovered gold particles. This is the reason why as-synthesized SiO<sub>2</sub>(BOS)/Au/TiO<sub>2</sub> with the lowest Si content could show some activity at relatively low temperatures ( $T_{50} = 141$  °C). Once the organic fragments were more sufficiently removed, the resulting catalysts exhibited improved thermal stability of gold particles, thus maintaining low T<sub>50</sub> values to some extent (table 1). Figure 6 shows the catalytic data of SiO<sub>2</sub>(APTMS)/Au/TiO<sub>2</sub> and SiO<sub>2</sub>(BOS)/Au/TiO<sub>2</sub>. The sintering-resistant capability of SiO<sub>2</sub>(APTMS)/Au/TiO<sub>2</sub> is similar to that of SiO<sub>2</sub>(TMOS)/Au/TiO<sub>2</sub> even though the Si content of the former is less than that of the latter. This could be attributed to the selective location of  $SiO_2$ through the interaction of amine groups with gold nanoparticles in the former catalyst system. The sintering-resistant ability of SiO<sub>2</sub>(BOS)/Au/TiO<sub>2</sub> is weaker than that of SiO<sub>2</sub>(TMOS)/Au/TiO<sub>2</sub>, in line with the lower surface decoration density of SiO<sub>2</sub> in the former case (Table 1).

# 3.4. Stability and regeneration of $SiO_2/Au/TiO_2$

The temporal stability seen on SiO<sub>2</sub>/Au/TiO<sub>2</sub> is acceptable, as demonstrated in figure 7. In this experiment, the reaction temperature was properly chosen to achieve an initial conversion below 100%, in order to clarify the stability on stream [10,51]. As shown in figure 7, the CO conversion on 500 °C-calcined SiO<sub>2</sub>(T-MOS)/Au/TiO<sub>2</sub> at 54 °C decreased from ca. 92 to 81% during the initial 6 h, and then gradually reached a steady-state value of ca. 70% in the following 18 h. In other experiments, the stability of 500 °C-calcined SiO<sub>2</sub>(APTMS)/Au/SiO<sub>2</sub> and SiO<sub>2</sub>(BOS)/Au/SiO<sub>2</sub> was found to be acceptable as well. For example, the initial CO conversion on 500 °C-calcined SiO<sub>2</sub>(BOS)/Au/TiO<sub>2</sub> at 53 °C was 97%, and the steady-state conversion was 89% when the reaction time was 24 h.



Figure 6. Panel A: CO light-off curves of as-synthesized SiO<sub>2</sub>(APT-MS)/Au/TiO<sub>2</sub> and those further calcined in O<sub>2</sub>-He at 500 °C for 1–5 h or at 700 °C for 1 h. Panel B: CO light-off curves of as-synthesized SiO<sub>2</sub>(BOS)/Au/TiO<sub>2</sub> and those further calcined in O<sub>2</sub>-He at 500 °C for 1–5 h or at 700 °C for 1 h.

The used catalyst can be regenerated. In a preliminary regeneration experiment, the used 500 °C-calcined SiO<sub>2</sub>(TMOS)/Au/TiO<sub>2</sub> catalyst, which had been stored for weeks in the U-type tube at ambient temperature after the stability test in figure 7, was treated in flowing O<sub>2</sub>-He at 400 °C for 1 h, and then cooled down to the reaction temperature. The initial conversion at 54 °C was still 91%, comparable to the 92% value achieved on the fresh catalyst. Thus, the deactivation of  $SiO_2/Au/TiO_2$  is not irreversible. The fact that the catalyst can be regenerated by calcination means that the catalyst deactivation is not due to the growth of gold particle sizes (because the deactivation caused by the growth of gold particle sizes is irreversible), but is likely due to the adsorption of surface intermediates such as carbonate [52]. Further research is in progress in our group to better understand the deactivation mechanism and regeneration methods.

# 3.5. Control experiments: comparison of SiO<sub>2</sub>/Au/TiO<sub>2</sub> and Au/SiO<sub>2</sub>/TiO<sub>2</sub>

Note that the current post-decoration design is different from the conventional one in which a sup-



Figure 7. CO conversion on 500 °C-calcined SiO<sub>2</sub>(TMOS)/Au/TiO<sub>2</sub> as a function of reaction time.

port (e.g.,  $SiO_2$ ) is modified by another metal oxide (e.g.,  $TiO_2$ ), and gold was subsequently loaded onto the modified support (e.g., TiO<sub>2</sub>/SiO<sub>2</sub>) [13,14]. The success in the deposition of gold onto  $TiO_2/SiO_2$  in the previous design lies in the fact that neat  $SiO_2$  is not suitable for deposition-precipitation of gold using HAuCl<sub>4</sub> precursor [6-10], because of the mismatch between the isoelectric point of SiO<sub>2</sub> (IEP  $\sim$  2) and the pH range needed to sufficiently hydrolyze the HAuCl<sub>4</sub> precursor to  $Au(OH)_3$  or  $Au(OH)_4^-$ (pH = 8-10) [10,13]. Therefore, the modification of low-IEP SiO<sub>2</sub> support with TiO<sub>2</sub> layer makes the resulting support more suitable for loading gold [13,14]. However, in the present case, if we modify the TiO<sub>2</sub> support with SiO<sub>2</sub>, and then load gold, the resulting SiO<sub>2</sub>/TiO<sub>2</sub> support is supposed to be less suitable for loading gold. Note that in a previous publication, the isoelectric point of TiO<sub>2</sub> was found to decrease from 6 to 2 after doping SiO<sub>2</sub> layer onto it [37]. Indeed, in control experiments, we found that the loading of gold onto such SiO<sub>2</sub>(TMOS)/TiO<sub>2</sub> support led to poor catalytic performance ( $T_{50} = 160$  and > 520 °C flowing heat treatment at 300 or 700 °C, respectively, figure 8). The reason is that gold was not effectively loaded onto SiO<sub>2</sub>/TiO<sub>2</sub>, as proven by the failure in detecting gold peaks by XRD (data not shown) and the low gold content as measured by ICP (0.1 wt%, table 1).

The results from our control experiment clearly indicate that the deposition-precipitation of gold is ineffective if the pre-formed  $SiO_2/TiO_2$  support is used to load gold. However, the deposition-precipitation will not be influenced if the decoration of  $SiO_2$  component is executed afterwards. The resulting  $SiO_2/Au/TiO_2$  catalyst combines both the high activity of Au-TiO<sub>2</sub> interface [6–10] and the high thermal stability of  $SiO_2$ component. The role of  $SiO_2$  is expected to be a textural promoter since the gold particle sizes are stabilized by the addition of  $SiO_2$  (table 1).



Figure 8. Comparison between the catalytic performance of  $SiO_2/Au/TiO_2$  and  $Au/SiO_2/TiO_2$ . In the first case, pre-formed  $Au/TiO_2$  was subsequently modified by  $SiO_2$  matrix (see also figure 3A). In the second situation for comparison,  $TiO_2$  was modified by  $SiO_2$  first, and then gold was loaded onto  $SiO_2/TiO_2$ .

# 3.6. Discussion and perspective in the context of the literature

To the best of our knowledge, this is the first report that post-decoration of active Au/TiO<sub>2</sub> by inert SiO<sub>2</sub> could mitigate the sintering of gold nanoparticles. We are aware of the work by Horváth et al. who post decorated inactive  $Au/SiO_2$  with active  $TiO_2$  patches to create new gold-TiO<sub>2</sub> boundaries, thus increasing the catalytic activity for CO oxidation [53]. Analogously, Guczi *et al.* post deposited active FeO<sub>x</sub> layers onto Au/  $SiO_2/Si(100)$  by pulsed laser deposition, and found that the activity of the  $FeO_x/Au/SiO_2/Si(100)$  model catalyst was higher than those of  $Au/SiO_2/Si(100)$  and  $FeO_x/$  $SiO_2/Si(100)$  because of the creation of gold-FeO<sub>x</sub> interface [54]. Others and us have post modified Au/ TiO<sub>2</sub> by nitrate [55], sulfate [56], and phosphate [57] to study the anion effects on activity. Lu and coworkers very recently reported that the impregnation of Au/C with KNO<sub>3</sub> could stabilize gold particles for selective CO oxidation in hydrogen-rich gas [58]. Going beyond the supported gold system, a recent communication by Kanazawa at Toyota Motor Corporation reported that the sintering of Pt particles on MFI zeolite was suppressed after impregnating Pt/MFI with tetramethyl orthosilane, and the resulting Pt catalyst could withstand thermal aging at 1000 °C [59]. Hence, it is expected that this post-decorating method may be extended to other supported metal systems for practical or commercial applications. However, due to the limited database, it is not clear whether other oxides can also stabilize metal particles.

Note that the chemical grafting method used in the current work is essentially solution-based [12,13,15,16,26]. We have chosen room temperature as the grafting temperature because our preliminary experiments indicated certain agglomeration of gold particles during chemical grafting in the presence of very

hot organic solutions containing the SiO<sub>2</sub> precursor (data not shown). 300 °C-calcined Au/TiO<sub>2</sub> was used instead of as-synthesized Au/TiO<sub>2</sub> for chemical grafting, due to the concern that supported Au<sup>3+</sup> may react with the ethanol solvent to form at least a proportion of big gold particles [29]. Further research opportunity may open up if gas-phase atomic layer deposition [60] is used instead of the solution-phase approach to post decorate metal catalysts. Nevertheless, there could be some issues related to the low vapor pressure, low reactivity, and limited availability of certain precursors, possible inhomogeneous coatings on power samples as opposed to flat substrates, as well as the influence of residual chloride (e.g., when using TiCl<sub>4</sub> as the precursor) on catalytic activity. In addition, one has to demonstrate the advantages of gas-phase atomic layer deposition over solution-based chemical grafting or conventional impregnation if the atomic-layer-decomposition method is to be highlighted. All of these are barely addressed in the conventional preparation of supported metal catalysts [61]. In addition, considering the fact that chemical grafting is able to be carried out in ultrahigh vacuum chambers [62,63], the post-decorating design may be extended to the construction and surface-science studies of a variety of model supported catalysts to be explored in the future [54].

# 4. Conclusions

In this work, we have developed a novel method to improve the thermal stability of gold catalysts. In this method, pre-formed Au/TiO<sub>2</sub> catalyst was decorated by amorphous SiO<sub>2</sub> matrix using tetramethyl orthosilane, (3-aminopropyl)trimethoxysilane, or tris(tert-butoxy)silanol as the precursor via a solution-based approach. Both the gold particle size and the activity in CO oxidation could be maintained to some extent even after several thermal aging. In contrast, Au/SiO<sub>2</sub>/TiO<sub>2</sub> prepared by the deposition-precipitation of gold complex onto SiO<sub>2</sub>-doped TiO<sub>2</sub> was not active for CO oxidation. The drastic difference in catalytic performance as a result of different synthesis sequence highlights the importance of rational catalyst design. Further research on the influence of synthesis conditions (e.g., synthesis temperature and Si content) and on the characterization of structural details is warranted.

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