The Electrical Breakdown of Thin Dielectric Elastomers: Thermal Effects

Zakaria Shamsul^a, Peter H. F. Morshuis^b, Benslimane Mohamed Yahia^c, Krist V. Gernaey^d, Anne Ladegaard Skov^{*a}

a: Danish Polymer Center, Department of Chemical and Biochemical Engineering, Technical University of Denmark, Søltofts Plads, Building 229, 2800 Kgs. Lyngby. b: Faculty of Electrical Engineering, Mathematics and Computer Science, Technology University of Delft, Mekelweg 4, 2628 CD Delft, The Netherland. c: Danfoss Polypower A/S, Nordborgvej 81 DK-6430 Nordborg. d: Center for Process Engineering and Technology, Department of Chemical and Biochemical Engineering, Technical University of Denmark, Søltofts Plads, Building 229, 2800 Kgs. Lyngby.

INTRODUCTION

Smart Structures/NDE

- > Dielectric elastomers are being developed for use in actuators, sensors and generators to be used in various applications, such as artificial eye lids¹, stretch sensors² and human motion energy generators³ [see fig 1].
- > For maximum efficiency, the devices are operated at high electrical fields which increases the likelihood for electrical breakdown significantly.
- > Thermal effects that may lead to electrical breakdown in thin PDMS film will be modeled. We assume the effect of temperature on electrical breakdown of thin PDMS film is different from polypropylene film as investigated by Xiaoguang et al. $(2003)^4$. > The main difference is that the PDMS elastomer is chemically crosslinked and thus the Young's modulus will not decrease with temperature as for the thermoplastic [see fig 2].



 $C\frac{dT}{dt} = \sigma(E,T)E^2 + \nabla(K(T)\nabla T)$

The heat balance equation where C= volumetric heat capacity, T= temperature, t= time, E= electrical field, ∇ = the Laplace operator, K(T) = the temperature dependent thermal conductivity, and $\sigma(E,T)$ = field and temperature dependent electrical conductivity.



MATERIALS AND METHODS

- > Several different types of silicone elastomers with different loadings of reinforcing silica particles as well as a permittivity enhancing filler (titanium dioxide) were studied.
- > Four of the elastomers are commercially available elastomers of either the type LSR (liquid silicone rubber) or RTV (room temperature vulcanizing).
- \triangleright A TA Instruments ARES G2 Rheometer was used to characterize the rheological properties of the prepared films.
- > Volume resistivity measurements were performed in a three-terminal cell by means of a Keithley 617 electrometer.
- > The thermogravimetric analysis (TGA) was performed with a TA Q500 equipped with autosampler.

RESULTS AND DISCUSSIONS



Elastosil LR 3043/30 film with 25 mm diameter and 0.8 mm thickness was used for characterization of the thermal dependence of the rheological properties of PDMS. The storage modulus and the loss tangent are plotted for temperatures between 25°C and 480°C.



dielectric properties function elevated The as of temperatures for several PDMS films: (A) Storage permittivity (B) Loss permittivity.

surface for 50 µm thick PDMS film.

Experimental Data

- \succ The increase of loss permittivity with increasing temperature is attributed to more dissipation of the electrical energy into heat⁵ [see fig 3].
- > The increase of electrical conductivity [see fig 4] at elevated temperature causes more heat production since the joule heating is directly proportional to the electrical conductivity⁶.
- > The TGA data [see fig 5] show that RT625 and XLR630 with filler, possess relatively low 2% degradation temperatures (around 300°C) whereas the other two have to be heated above 400°C before significant degradation takes place.

Numerical Electrothermal Prediction of Breakdown

- > The electrothermal breakdown was modeled based on numerical prediction as described in [see fig 6].
- > In solid dielectrics, electrical breakdown may be thermal which means it is caused by the fact that heat generated within the film cannot be dissipated sufficiently and thereby leads to thermal instability⁶ [see fig 7].







(M/µm)

CONCLUSION

- > The effect of temperature on dielectric properties of different systems of PDMS dielectric elastomers has been studied experimentally and a model of electrothermal breakdown in thin PDMS based dielectric elastomers has been applied.
- > From both methods, it can be concluded that electrothermal breakdown of the materials is strongly influenced by the increase in both dielectric permittivity and conductivity.
- > The electrothermal breakdown may not be a major factor to cause electrical breakdown in thin PDMS based dielectric elastomers since the required electrical field required for thermal runaway is about 5 times larger than the reported breakdown fields of silicones.

REFERENCES

[1] S. H. Goodwin-Johansson, P. H. Holloway, G. McGuire, L. J. Buckley, R. F. Cozzens, R. W. Schwartz, and G. J. Exarhos, "Artificial eyelid for protection of optical sensors," Proceeding SPIE 3987, Smart Structures and Materials 2000: Electroactive Polymer Actuators and Devices (EAPAD), 225-231 (2000).

- > The general behavior of temperature versus electric field for PDMS film for which the temperature at the center of the sample just before thermal runway is only a few degrees above the boundary temperature [see fig 8]. With only a few volts increase across the sample, thermal runaway occurs very rapidly.
- > The breakdown field as function of electrical conductivity for 50 µm thick PDMS film [see fig 9]. The plot illustrates that higher electrical conductivity causes a lower breakdown field.
- > The breakdown field exhibits a hyperbolic decrease as the position of x is closer to the film surface as the result of the heat generated inside the film can be removed rapidly to the surrounding [see fig 10].

[2] Anonymous, "PolyPower stretch sensors in action" http://www.polypower.com/products /sensors/Videos.htm (17 February 2014). http://www.polypower.com.

[3] R. D. Kornbluh, E. Joseph, and M.C. Brian. "A scalable solution to harvest kinetic energy." SPIE Newsroom (2011).

[4] Q. Xiaoguang, Z. Zhong, and S. Boggs, "Computation of electro-thermal breakdown of polymer films," in Electr. Insul. Dielectr. Phenomena, 2003. Annu. Report. Conf., pp. 337-340 (2003).

[5] A. R. Von Hippel and A. R. Hippel, [Dielectrics and waves], Artech House (1954).

[6] J. C. Fothergill and L. A. Dissado, [Electrical degradation and breakdown in polymers], IET (1992).

ACKNOWLEDGEMENT

The authors gratefully acknowledge the financial support from the Ministry of Education of Malaysia and Universiti Malaysia Pahang. Danfoss PolyPower A/S is also acknowledged for financial support to Z. Shamsul.

*Contact author: al@kt.dtu.dk; phone +45 4525 2825